

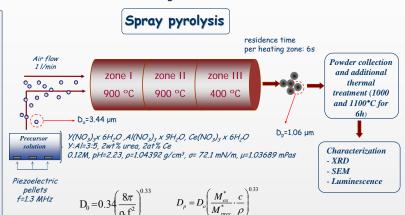
# UREA-ASSISTED SELF-COMBUSTION AEROSOL SYNTHESIS OF Y3AI5O12:Ce3+

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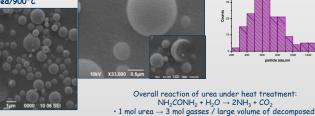
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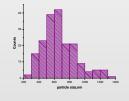
## **Abstract**

A simple route to pure garnet phase synthesis is difficult to achieve, due to the formation of other stable phases of Y2O3-Al2O3 system (monoclinic, perovskite and hexagonal). In this study, ureaassisted spray pyrolysis was employed, in the effort to synthesize a pure YAG phase with good structural and morphological characteristics. This route can be regarded as a self-combustion synthesis confined within a droplet where urea acts as an in-situ source of thermal energy due to its decomposition. The process involved ultrasonic aerosol formation (1.3 MHz) from the ureamodified nitrate precursor solution and consequent control over the aerosol decomposition united with self-combustion of droplets within a high-temperature tubular flow reactor (900°C). The as-prepared amorphous particles were additionally thermally treated in air at 1000 and  $1100^{\circ}\text{C}$  for 6h. Morphological features were investigated by means of SEM analysis while XRD analysis gave insight into the structural properties. Obtained data were refined by Rietveld method. Optical properties were investigated by photoluminescent measurements.



# SEM as-prepared/900°C



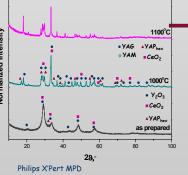


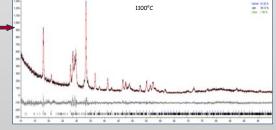
As-prepared particles are nonagglomerated with moderately broad size distribution. Smooth particle surface is in agreement with amorphous phase presence confirmed by XRD. Insight into a broken particle leads to the presumption that volume precipitation occurs.

Thermal treatment effect on particle morphology is obvious since there is an increase in particle surface roughness with temperature due to promoted crystallization.

SEM JEOL JSM-6390-LV

#### **XRD**





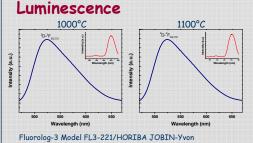
gasses produces internal pressure inside the particles

Temperature	1100°C		
Phase name	YAG	YAM	CeO <sub>2</sub>
Phase content, wt%	61.92(9)	36.12(9)	1.96(2)
Space group	Ia-3d, cubic	P21/c, monoclinic	Fm-3m, cubic
Lattice parameters	a=12.035(7)	a=7.737(3) b=10.537(6) c=11.731(5) β=107.37(4)	a=5.412(1)
Crystallite size, nm	80.5 (9)	14.2(5)	57.0 (9)
Microstrain, Δα/α %	0.592(5)		
R <sub>Bragg</sub> ,%	1.966	2,017	2,420
gof	1,134		

Thermal treatment leads not only to the segregation of targeted  $Y_3AI_5O_{12}$  phase but also to the formation of other phases as-prepared: amorphous phase present and the rising peaks are attributed to  $Y_2O_3$  (PDF 89-5591),  $CeO_2$  (PDF 65-5923) and hexagonal YAIO3 phase space group P63/mmc (PDF 74-1334)

1000°C: beside YAG phase monoclinic YAM, space group P21/c (PDF 34-0368) and hexagonal YAP phases are present CeO2 forms as well

1100°C: Rietveld refinement → the major phase content is attributed to YAG phase cubic structure, space group Ia-3d (PDF 33-0040)



Broad green-yellow emission band in the range of 470-670 nm peaking at 521nm is ascribed to the electron transition from the excited state of  $^2D_J$  to the ground states of  $^2F_{5/2,7/2}$  of  $Ce^{3+}$  ions in the  $Y_3Al_5O_{12}$  host material.

Xenon lamp 450W: Aex=343 nm

## Conclusion

Urea-assisted aerosol synthesis of YAG:Ce3+ led to formation of fine powders with good morphological features i.e. spherical, nonagglomerated and submicron particles were synthesized. However, XRD analysis reviled that the phase segregation has occurred and that YAG phase is followed with the presence of hexagonal and monoclinic intermediate phases of Y2O3-Al2O3 Nevertheless, the system. luminescent measurements revile typical Ce3+ ion emission from the  $Y_3AI_5O_{12}$  host lattice and prove Ce3+ incorporation into the host matrix.

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